

## Vanadium-doped nanocrystalline TiO<sub>2</sub> layer prepared by spin coating pyrolysis

KYU-SEOG HWANG\*

*Department of Applied Optics and Institute of Photoelectronic Technology, Nambu University, 864-1 Wolgye-dong, Gwangsan-gu, Gwangju 506-824, Republic of Korea*  
E-mail: khwang@nambu.ac.kr

JUN-HYUNG AHN, BYUNG-HOON KIM

*Department of Materials Science and Engineering, Chonnam National University, 300 Yongbong-dong, Buk-gu, Gwangju 500-757, Republic of Korea*

**Published online:** 10 March 2006

TiO<sub>2</sub> thin films have received great attention in recent years because of their excellent optical and electrical properties [1–3]. Roles of electron-hole pairs generated by the ultraviolet (UV) light irradiation have been studied in the basic and application field. The band gap energy requires that near-UV light be used to photoactivate this very attractive photocatalyst. Unfortunately, in solar energy applications only ~3 % of the solar light is absorbed. It would be advantageous, therefore, if TiO<sub>2</sub> could be photosensitized by visible light.

Many studies have been devoted to the extension of the photoresponse and improvement of the photoactivity by dye-sensitizing, doping, surface modification, and depositing noble metal [4–6]. Doping with metal cations has been attempted for photo-catalytic applications by shifting the threshold for photonic excitation of the TiO<sub>2</sub> towards the visible. All metal dopants are conveniently substituted into the TiO<sub>2</sub> lattice, if their ionic radii are identical or nearly identical to that of the Ti<sup>4+</sup>. These systems are being prepared in the form of thin films due to the advantages of their use as electrodes, transparent coatings, etc. [7, 8].

Vanadium (V) doping increases carrier lifetime and apparently also extends the absorption range of TiO<sub>2</sub> [7]. Herein, we examine the crystallographic and spectroscopic behavior of V-doped TiO<sub>2</sub> thin films prepared by spin coating-pyrolysis process. Metal naphthenate precursor was used as a starting material because of stability in air and ease of handling.

Ti- and V-naphthenates (Nihon Kagaku Sangyo Co. Ltd., Japan) were used as a precursor in the spin coating-pyrolysis process and by diluting the sol with toluene (molar ratio of V : Ti = 9 : 95). The concentration of metal ions in coating solution was about 4 wt.%.

Thin films were fabricated by spin coating onto 2.5 cm × 2.5 cm × 1 mm substrates of the soda lime silica

glass [SLSG, Corning 2947, O (60 at.%), Si(24.5 at.%), Na (10 at.%), Ca (2.5 at.%), Mg (2 at.%), Al (1 at.%)] at a rotation speed of 1500 rpm for 10 s. After each deposition, the coating film was pyrolyzed in air at 500 °C for 10 min to decompose organic species. For multiple coatings, the above-mentioned processes were repeated five times to obtain the resulting films with a thickness of approximately 0.6–0.7 μm. The resulting films were directly annealed into a preheated furnace with temperature at 550 °C for 30 min in air.

The phase evolution of the films was examined by an high resolution X-ray diffraction (HRXRD, X'pert-PRO, Philips, Netherlands) using a CuK<sub>α</sub> radiation. The thickness of the films was measured by observations of fractured cross-section using a field emission-scanning electron microscope (FE-SEM, S-4700, Hitachi Co., Japan). The microstructure of the films after annealing was examined using a FE-SEM. Spectroscopic analysis of the films was performed by using a UV-Visible-NIR spectrophotometer (Cary 500 Scan, Varian Co., Australia, its wavelength range is 300–900 nm).

Fig. 1 shows the XRD pattern of the film after annealing at 550 °C. An anatase phase with a strong (110) reflection ( $2\theta = 25\text{--}26^\circ$ ) appeared when the temperature reached 550 °C, although the films after preheating at 500 °C for 10 min exhibit amorphous character, not shown here. There was no peaks for rutile of TiO<sub>2</sub>, vanadium oxide, and vanadium titanate. Of the two crystal forms of TiO<sub>2</sub> catalysts, the anatase invariably exhibits a higher photoactivity than rutile does [9]. Therefore, the higher the anatase-to-rutile ratio the crystal has, the more photoactive the catalyst is. Metal dopants, such as Fe and Cr, would be increasing oxygen vacancies in TiO<sub>2</sub> crystal lattice. These oxygen vacancies are responsible for the enhancement of transformation from anatase to rutile [10]. Rutilisation was already induced at 400 °C in Fe-doped TiO<sub>2</sub> sample

\* Author to whom all correspondence should be addressed.

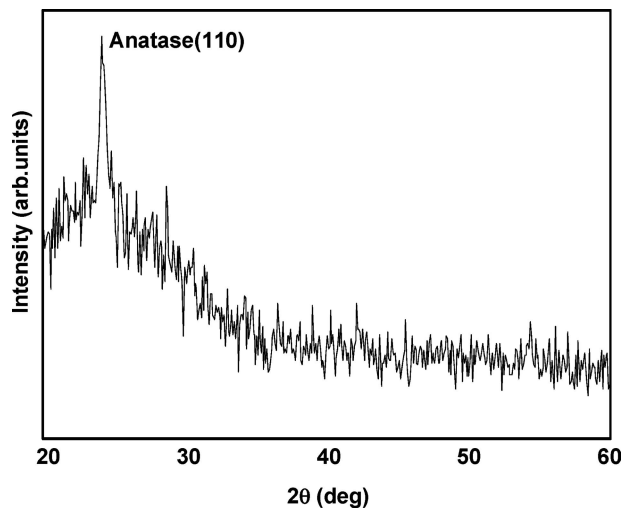


Figure 1 XRD spectra of the 5 mol% V-doped TiO<sub>2</sub> thin film on SLSG substrate after annealing at 550 °C.

[5]. This would be for the acceptor dopants. However, in this work, V-doped TiO<sub>2</sub> thin films fully crystallized into anatase structure at high annealing temperature, 550 °C. We assume that V would act as a donor dopant in TiO<sub>2</sub> lattice [11].

Fig. 2 shows surface microstructure (a) and fractured cross-section (b) of the film after annealing at 550 °C. The film exhibits a dense microstructure with a nano grain size, as shown in Fig. 2a. No evidently aggregated particles are present. The fractured cross-section of the layer of about 0.6–0.7 μm thickness appears dense and uniform.

Fig. 3 shows the visible spectra in the wavelength range 300–900 nm of the film on SLSG substrate after annealing at 550 °C. Relative high transmittance at visible range and a clear absorption edge of the film were observed. The high transmittance of the film is attributed to the small particle size which eliminate light scattering [12]. From this spectrum, it is apparent that films present relatively high optical quality, with an absorption in the visible region that is characterized by the typical interference pattern found when a transparent thin film is deposited onto a substrate of different refractive index. Furthermore, the

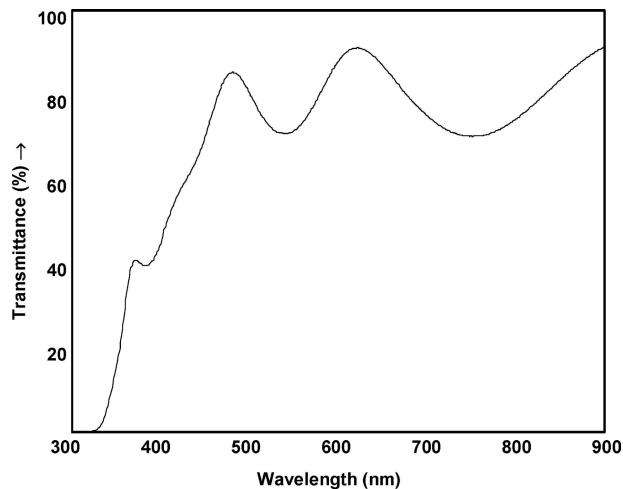


Figure 3 Transmittance in visible wavelength range of the 5 mol% V-doped TiO<sub>2</sub> thin film on SLSG substrate after annealing at 550 °C.

film containing V showed a shift towards the visible in the absorption threshold.

Furthermore, we calculated the band gap values of the films from their transmission spectra using the following equation [13]:

$$\alpha = (h\nu - E_g)^{1/2}$$

where  $\alpha$  is the absorption coefficient,  $E_g$  is the band gap and  $h\nu$  is the photon energy. The intersection of the linear region on the  $h\nu$  axis of the plot of  $\alpha^2$  versus  $h\nu$  gives the band gap of the ZnO films deposited on SLSG substrates. As shown in Fig. 3, using this method [14] the band gap value of the films is found to be 3.41 eV. The estimated value of the band gap for the present film, V-doped TiO<sub>2</sub>, is much smaller than that of previous TiO<sub>2</sub> film (3.70 eV) [15]. Generally, the film consisting of fine crystallites show ‘blue shift’ [16]. A comparison of optical energy gaps between pure TiO<sub>2</sub> and V-doped TiO<sub>2</sub>, thin films shows an obvious red shift in V-doped TiO<sub>2</sub> films.

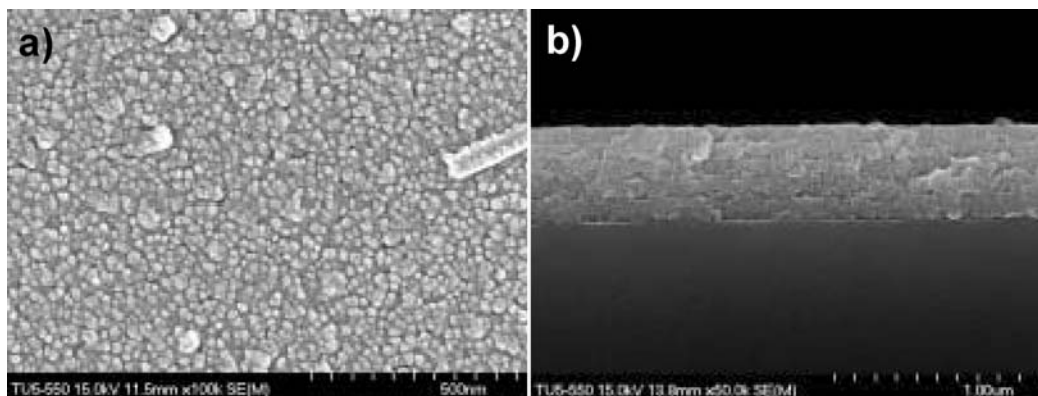


Figure 2 FE-SEM photographs of the surface morphology (a) and fracture cross-section (b) of the 5 mol% V-doped TiO<sub>2</sub> thin film on SLSG substrate after annealing at 550 °C.

In summary, V-doped TiO<sub>2</sub> thin films were prepared on SLSG substrates at 550 °C from metal naphthenates precursor. Only anatase phase with a strong (110) reflection appeared when the temperature at 550 °C. The film exhibits a dense microstructure with a nano grain size by FE-SEM observation. The estimated value of the band gap for the V-doped TiO<sub>2</sub>, is much smaller than that of previous pure TiO<sub>2</sub> film (3.67 eV).

## References

1. C. C. TING and S. Y. CHEN, *J. Mater. Res.* **16**(6) (2001) 1712.
2. A. H. C. CHAN, J. F. PORTER, J. P. BARFORD and C. K. CHAN, *ibid.* **17**(7) (2002) 1758.
3. M. NAKAMURA, T. AOKI, Y. HATANAKA, D. KORZEC and J. ENGEMANN, *ibid.* **16**(2) (2001) 621.
4. G. H. LI, L. YANG, Y. X. JIN and L. D. ZHANG, *Thin Solid Films* **368** (2000) 163.
5. F. GRACIA, J. P. HOLGADO, F. YUBERO and A. R. GONZÁLEZ-ELIPE, *Surf. Coat. Tech.* **158/159** (2002) 552.
6. H. KIM, N. HARA and K. SUGIMOTO, *J. Electrochem. Soc.* **146**(3) (1999) 955.
7. N. SERPONE, D. LAWLESS, J. DISDIER and J. M. HERRMANN, *Langmuir* **10** (1994) 643.
8. F. C. GENNARI and D. M. PASQUEVICH, *J. Mater. Sci.* **33** (1998) 1571.
9. K. KATO, A. TSUZUKI, H. TAODA, Y. TORII, T. KATO and Y. BUTSUGAN, *ibid.* **29** (1994) 5911.
10. S. RIYAS, V. A. YASIR and P. N. M. DAS, *Bull. Mater. Sci.* **25**(4) (2002) 267.
11. S. KARVINEN, *Solid State Sci.* **5**(5) (2003) 811.
12. H. W. RYU, J. S. PARK, K. J. HONG, K. S. HWANG, B. A. KANG and J. S. OH, *Surf. Coat. Tech.* **173** (2003) 19.
13. K. PARK, D. MA and K. KIM, *Thin Solid Films* **305** (1997) 201.
14. D. SHIMONO, S. TANAKA, T. TORIKAI, T. WATARI and M. MURANO, *J. Ceram. Proc. Res.* **2**(4) (2001) 184.
15. J. H. JEONG, J. H. AHN, B. H. KIM, Y. S. HEON, K. O. JEON and K. S. HWANG, *J. Kor. Phys. Soc.* **46**(2) (2005) 559.
16. T. WANG, H. WANG, P. XU, X. ZHAO, Y. LIU and S. CHAO, *Thin Solid Films* **334** (1998) 103.

*Received 9 August  
and accepted 15 August 2005*